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Enhancing photodetection of WSe₂ phototransistors by reducing charge scattering with hBN passivation

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Abstract

Two-dimensional (2D) semiconductors like tungsten diselenide (WSe₂) have significant potential for next-generation phototransistors because of their broad detection range and strong light-matter interaction. However, dielectric disorder at the interface with conventional silicon dioxide (SiO₂) substrates hampers the device performance by introducing surface charge traps and oxygen dangling bonds, which degrade carrier transport and increase low-frequency noise (LFN). Here, we show that simply incorporating a hexagonal boron nitride (hBN) passivation layer between WSe2 and SiO2 effectively reduces interfacial disorder by creating a clean van der Waals interface through optoelectrical and LFN analysis. The WSe₂/hBN phototransistors demonstrate a 100-fold decrease in LFN, a 100-fold increase in responsivity, and a 10-fold improvement in specific detectivity compared to WSe₂ devices without an hBN layer. This enhancement in device performance is ascribed to shielding Coulomb scattering and maintaining the intrinsic transport properties of WSe₂. Our findings underscore the importance of passivating the channel scattering sources of 2D-based phototransistors in enhancing photodetection performance.

1. Introduction

Two-dimensional (2D) semiconductors, like transition metal dichalcogenides (TMDs), are considered next-generation semiconductors due to their potential for overcoming the bulk materials' thickness limits [1–4]. Specifically, TMDs are promising phototransistor materials due to their strong light-matter interaction, wide detection range, and interesting photonic phenomena [5-13]. However, a major drawback of low-dimensional materials, including TMDs, is their susceptibility to dielectric disorder, which dominates the optical and electronic properties of devices compared to bulk materials [14, 15].

The major feature of the dielectric disorder is the Coulomb scattering effect during carrier transport, originating from a variety of impurities of the dielectric and the interface [1, 7, 16]. Coulomb scattering leads to a decline in device mobility, a reduction in the current level, and an increase in the device's low-frequency noise (LFN) [15–18]. Since reduced current and increased LFN degrade the performance of phototransistors, minimizing Coulomb scattering induced by the dielectric is essential [19, 20]. Changing the dielectric to high-k or ferroelectric materials is a good solution to address these issues, due to their shielding effect against the scattering sources [21-24]. However, high-k dielectric can induce serious surface optical phonons, and ferroelectric dielectric can induce unintended electrostatic doping due to polarization effects [25–27]. Since then, dielectrics with almost zero dangling bonds, like hexagonal boron nitride (hBN), have been employed to mitigate the dielectric disorder [28, 29]. Utilizing hBN is a great solution to dielectric disorder, owing to its clean interface with TMDs and the absence of phonon-related penalties.

In our study, we reduced LFN and preserved the intrinsic carrier transport properties for enhancing the photodetection performance of tungsten diselenide (WSe₂) phototransistors by separating WSe₂ and amorphous silicon dioxide (SiO₂) using hBN. The hBN, a well-known material that acts as a passivation layer, effectively suppressed the dielectric disorder by forming a van der Waals gap with the WSe₂ channel. The hBN layer improved the overall photodetection performance of WSe₂/hBN devices by a 100-fold reduction in LFN, a 100-fold improvement in responsivity, and a 10-fold enhancement in specific detectivity compared to WSe₂ devices without hBN. This study highlights that passivating the scattering sites of the channel-dielectric interface of the 2D-based phototransistors using dangling-bond-free materials is effective in enhancing photodetection performance.

2. Experiment

The hBN and WSe₂ flakes (both purchased from HQ Graphene) were mechanically exfoliated from bulk crystals and transferred onto a 270 nm $SiO_2/p++$ Si substrate. Using an optical microscope and an atomic force microscopy (AFM) system (NX-10, Park Systems), thick hBN and few-layer WSe₂ flakes were selected. For WSe₂/hBN heterostructures, a selected few-layer WSe₂ flakes were layered on top of the hBN flakes using the stamping technique with a manipulator (AP-4200, Unitek). Here, we used propylene carbonate-coated polydimethylsiloxane stamps for improved adhesion with WSe₂. After preparing the channel layers, we spin-coated the flakes with methyl methacrylate and poly methyl methacrylate to create a double-electroresist layer. Source and drain electrodes were patterned using an electron-beam lithography system (JSM-6510, JEOL), and a 40 nm layer of gold was deposited using an electron-beam evaporator (KVE2004, Korea Vacuum Tech). Completed phototransistors were heated at 200 °C for 2 h in an argon environment via a rapid thermal annealing system (KVR-4000, Korea Vacuum Tech) to enhance their electrical properties. More details on device fabrication can be found in figure S1 of the supplementary information.

The electrical characteristics and LFN analysis of the WSe_2/hBN devices were measured with a probe station (M6VC, MSTECH) and a semiconductor parameter analyzer (Keithley 4200). The photodetection performance was measured under constant laser illumination at a wavelength of 532 nm with varying powers. The laser beam illuminated the phototransistors globally, with a size specified at a few millimeters in diameter. All characterizations were conducted at room temperature and in a vacuum to eliminate the effects of extraneous variables such as humidity or other gases present in the air. Also, all devices were stored in a vacuum desiccator.

3. Results and discussion

The WSe₂/hBN devices were observed via optical characterization. The optical image of a WSe₂/hBN device can be seen in figure 1(a). The thickness of hBN and WSe₂ flakes was measured using an AFM (figure 1(b)). We employed thick hBN over 30 nm and a few-layer WSe₂ near 15 nm (supplementary figure S2). Figures 1(c) and (d) are the photoluminescence (PL) and the Raman mapping images of the WSe₂/hBN device, illustrating that the WSe₂/hBN heterostructure was successfully constructed. For PL measurements, the hBN displayed deep ultraviolet emission, which fell outside the measurement range, whereas the WSe₂ flake exhibited a prominent A peak (\sim 780 nm) [30, 31]. Additionally, the E_{2g} peak (\sim 1365 cm⁻¹) of the hBN and the E^{1}_{2g} peak (\sim 248 cm⁻¹) of the WSe₂ were clearly shown in Raman spectroscopy [30, 32]. Figure 1(e) presents a schematic of the phototransistor experimental setup. A 532 nm laser was used and modulated with a chopper. The device current was measured using a semiconductor parameter analyzer while applying gate and drain voltages. Subsequently, the optoelectrical properties of the device, including LFN analysis and photodetection performance calculations, were evaluated.

Figure 2(a) shows the transfer characteristics (drain current versus gate voltage, $I_{\rm D}-V_{\rm G}$) of representative WSe₂ and WSe₂/hBN devices. Both the WSe₂ and WSe₂/hBN devices exhibited ambipolar behaviors; however, the n-type properties of the WSe₂/hBN devices were prominent compared to the WSe₂ devices. The gate leakage current was sufficiently small to be considered negligible in further analysis (supplementary figure S3). Also, the output characteristics (drain current versus drain voltage, $I_{\rm D}-V_{\rm D}$) of the WSe₂ and WSe₂/hBN devices showed that the WSe₂/hBN device's drain current was significantly higher than that of the WSe₂ device, as shown in figure 2(b). This trend was consistently observed in other devices as well (supplementary figure S4). These indicate that the insertion of hBN layer between the WSe₂ flake and SiO₂ substrate can enhance the electron transport properties of a transistor. In addition to modifying electrical properties, hBN also played a critical role in determining the LFN characteristics of the device. Figure 2(c) shows the device's normalized noise (normalized power spectral density with drain current). The LFN of the WSe₂-based devices exhibited a 1/f noise-like characteristic since the 1/f noise is the primary noise source of 2D TMDs [18, 33, 34]. For more details about the LFN analysis, see figure S5 of the

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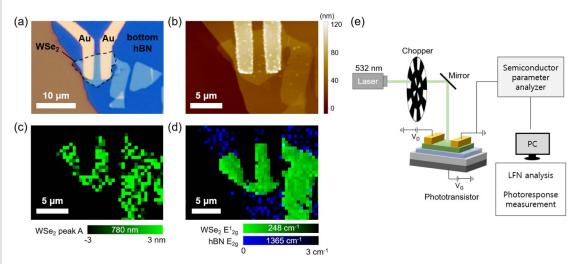


Figure 1. (a) The optical image, (b) the AFM image, (c) the PL mapping image, and (d) the Raman mapping image of the WSe₂/hBN device. (e) The schematic of the phototransistor experimental setup.

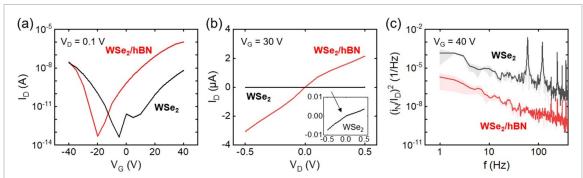


Figure 2. (a) The transfer characteristics for the WSe₂ and WSe₂/hBN devices at $V_D=0.1$ V. (b) The output characteristics for the WSe₂ and WSe₂/hBN devices at $V_G=30$ V. (c) The normalized noise of the WSe₂ and WSe₂/hBN devices at $V_G=40$ V.

supplementary information. The LFN was 100 times lower in the WSe₂/hBN device than in the WSe₂ device, which shows that hBN can effectively suppress the LFN. Additionally, LFN reduction was consistent across various gate voltages (figure S6 of the supplementary information). Note that contact configuration was consistently maintained as gold-top contacts for all devices, so it is reasonable to attribute the observed reduction in LFN to the hBN passivation effect.

The enhanced electron transport and reduced LFN can be attributed to the passivation effect of hBN, which mitigates the impact of defects such as oxygen dangling bonds and surface charge traps typically present on SiO₂ dielectrics. Oxygen dangling bonds are known to act as p-type dopants in the WSe₂ channel, while surface charge traps contribute to increased LFN and decreased carrier transport properties [14, 16]. Introducing an hBN passivation layer thus leads to improved electrical performance and LFN suppression.

Figure 3(a) shows that the WSe₂ and the WSe₂/hBN devices exhibited clear photoresponse. In addition, the current of the WSe₂/hBN device was much more stable than that of WSe₂ alone, indicating that the WSe₂/hBN device generated less LFN. Likewise, in other devices, WSe₂/hBN devices consistently demonstrated lower current fluctuations (supplementary figure S7). Both devices exhibited very fast photoresponse of a few milliseconds, but the WSe₂/hBN device showed quicker response time (supplementary figure S8). The more comparisons of optoelectronic performances between the WSe₂ and WSe₂/hBN devices are shown in figures 3(b) and (c). Figure 3(b) plots the responsivity ($R = I_{ph}/P_{eff}$, where I_{ph} is the photocurrent and I_{eff} is the effective incident power to the channel), which is the power-to-photocurrent conversion ratio [19, 20]. As shown in figure 3(b), it can be observed that regardless of the power of the laser, the responsivity of the WSe₂/hBN device was always 100 times higher than that of the WSe₂ device. Figure 3(c) further solidifies the WSe₂/hBN device's superior optoelectronic properties by plotting the specific detectivity (I_{eff}) and I_{eff}), where I_{eff} is the area of the device, I_{eff} is the bandwidth, and I_{eff} is the noise current), which is the sensitivity of the device [19, 20]. We calculated the specific detectivities based on the I_{eff} noise using LFN analysis for an accurate evaluation. Note that if we consider shot noise to be

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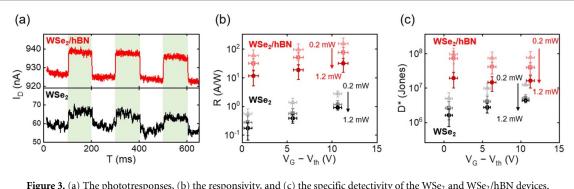


Figure 3. (a) The phototresponses, (b) the responsivity, and (c) the specific detectivity of the WSe2 and WSe2/hBN devices.

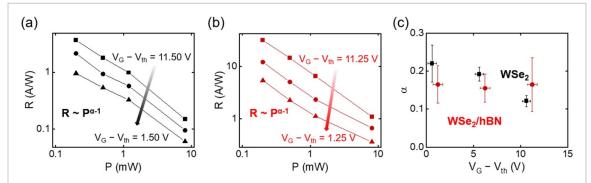
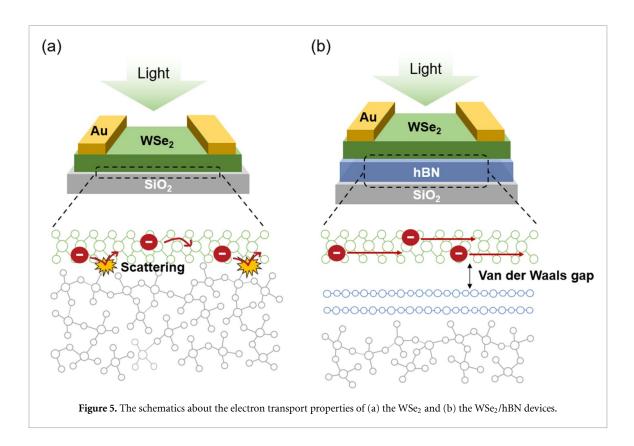


Figure 4. The responsivity of (a) the WSe2 and (b) WSe2/hBN devices fitted to a power law for incident power. (c) The gate voltage dependence of the α .

the only primary noise source, the specific detectivity is overstated (supplementary figure S9) [18]. The WSe₂/hBN device displayed a 10-fold improvement in specific detectivity over the WSe₂ device due to its lower LFN and higher responsivity.

The photogating effect is a plausible contributor to enhanced photodetection performance. As one of the primary photogeneration mechanisms in 2D TMDs, the photogating effect has been well known to enhance photocurrent and photodetection performance through channel gating [35–37]. Its influence can be evaluated based on the power-law relationship between responsivity and incident power $(R \sim P^{\alpha-1})$, where a smaller value of α in the expression indicates a dominant photographic contribution. The α value reflects the degree of the photogating effect under different gate voltages. Figures 4(a) and (b) present the responsivity of the WSe₂ and WSe₂/hBN devices, fitted using the power-law relation to extract α values. As shown in figure 4(c), the contribution of the photogating effect was nearly identical in both types of devices. This suggests that applying the hBN layer had minimal impact on the photogeneration mechanisms and that the improvement in photodetection performance was attributed to other factors, like reduced Coulomb scattering and suppressed defects.

Figure 5 illustrates the mechanisms of how hBN enhances photodetection performance. The amorphous nature of SiO₂ results in a high density of surface defects, leading to a poorly defined interface upon integration with WSe₂, characterized by numerous oxygen dangling bonds and surface charge traps that can trap electrons from the neighboring channel and introduce electrical noise (figure 5(a)) [38]. In contrast, hBN forms a clean interface with WSe₂ through van der Waals interaction and physically separates them from the underlying SiO₂ [39, 40]. Using a passivation layer effectively reduces the WSe₂ channel from Coulomb scattering induced by surface charges on SiO₂, thereby improving electron transport (figure 5(b)). When the light is illuminated on the WSe₂/hBN device, the photogenerated carriers can flow more efficiently, resulting in enhanced photocurrent. Additionally, the effect of surface traps and Coulomb scattering is minimized, leading to reduced LFN. As a result, the suppression of LFN and increased photocurrent directly translated into enhanced photodetection performance. The improvement is attributed to the passivation provided by hBN; therefore, the enhancement can be universally observed under laser illumination conditions that excite carriers in WSe₂.



4. Conclusion

In this study, we demonstrated that passivating the scattering sites of SiO₂ dielectric using hBN effectively suppresses dielectric disorder and enhances the photodetection performance of WSe₂ phototransistors. Optoelectrical and LFN analysis showed that our WSe₂/hBN devices exhibited near a 100-fold enhancement in responsivity and LFN reduction and a 10-fold increase in specific detectivity compared to pristine WSe₂ devices. These improvements are attributed to forming a clean van der Waals interface, which minimizes interfacial charge scattering and preserves the intrinsic electronic properties of the WSe₂ channel. These findings underscore the critical role of dielectric engineering in 2D optoelectronic devices and highlight dangling-bond-free materials as effective passivation materials for suppressing interfacial disorder and enhancing the performance of optoelectronic devices.

Data availability statement

All data that support the findings of this study are included within the article (and any supplementary files).

Acknowledgments

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