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# Tailoring the Electrical Characteristics of MoS<sub>2</sub> FETs through Controllable Surface Charge Transfer Doping Using Selective Inkjet Printing

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**ABSTRACT:** Surface charge transfer doping (SCTD) has been regarded as an effective approach to tailor the electrical characteristics of atomically thin transition metal dichalcogenides (TMDs) in a nondestructive manner due to their two-dimensional nature. However, the difficulty of achieving rationally controlled SCTD on TMDs *via* conventional doping methods, such as solution immersion and dopant vaporization, has impeded the realization of practical optoelectronic and electronic devices. Here, we demonstrate controllable SCTD of molybdenum disulfide (MoS<sub>2</sub>) field-effect transistors using



inkjet-printed benzyl viologen (BV) as an n-type dopant. By adjusting the BV concentration and the areal coverage of inkjet-printed BV dopants, controllable SCTD results in BV-doped  $MoS_2$  FETs with elaborately tailored electrical performance. Specifically, the suggested solvent system creates well-defined droplets of BV ink having a volume of ~2 pL, which allows the high spatial selectivity of SCTD onto the  $MoS_2$  channels by depositing the BV dopant on demand. Our inkjet-printed SCTD method provides a feasible solution for achieving controllable doping to modulate the electrical and optical performances of TMD-based devices.

KEYWORDS: molybdenum disulfide, chemical doping, surface charge transfer doping, inkjet printing, field-effect transistor

tomically thin two-dimensional (2D) transition metal dichalcogenides (TMDs) have emerged as building blocks for implementing next-generation nanoelectronics.<sup>1-3</sup> Among TMDs, layered molybdenum disulfide  $(MoS_2)$  has gained great attention due to its high carrier mobility, chemical and thermal stability, absence of dangling bonds, and tunable electronic energy band depending on the number of layers.<sup>4-6</sup> Specifically, chemical vapor deposition (CVD)-grown monolayer MoS<sub>2</sub> offers promising opportunities to realize large-area functional applications such as field-effect transistors (FETs) and optoelectronic devices.<sup>7-10</sup> For practical applications beyond laboratory-scale devices, the capability to tune the electrical characteristics of MoS<sub>2</sub> is important in building integrated logic circuits.<sup>11-13</sup> For mature three-dimensional (3D) Si-based semiconductor devices, thermal diffusion and ion implantation doping have been widely employed to facilitate the rational modulation of electrical characteristics.<sup>13,14</sup> However, these approaches have limitations when introduced to 2D TMDs because they can

result in undesirable lattice distortion, a substantial number of defects, and doping instability.<sup>14,15</sup> Therefore, it is necessary to develop appropriate doping strategies for TMDs to modulate their electrical characteristics without deteriorating their intrinsic properties.

Recently, surface charge transfer doping (SCTD), which can induce electron donation or withdrawal between the deposited dopants and host TMDs, has been intensively investigated.<sup>16–29</sup> The carrier concentration of TMDs can be controlled by the SCTD without the lattice distortion of host TMDs. Various organic dopants have been reported to conduct SCTD on TMDs, such as tetrafluoro-

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Figure 1. (a) Schematic of the MoS<sub>2</sub> FETs with inkjet-printed Ag contacts. (b) Optical images of the  $10 \times 10 \text{ MoS}_2$  FET array. Inset image shows a single MoS<sub>2</sub> FET. The scale bar is 200  $\mu$ m. (c) Histogram of the field-effect mobility distribution of the MoS<sub>2</sub> FETs. (d)  $I_{DS}-V_{GS}$  curves and (e)  $I_{DS}-V_{DS}$  of the transferred monolayer MoS<sub>2</sub> FETs. (f) Logarithmic scale plot for  $I_{DS}-V_{DS}$  of MoS<sub>2</sub> FETs with an average  $\gamma$  value. (g) Raman spectra showing the  $E_{2g}^{-1}$  and  $A_{1g}$  modes, (h) PL spectra containing the A exciton peak, and (i) PL mapping image containing the A exciton peak at 1.84 eV. The inset indicates the optical image of the measured MoS<sub>2</sub> FET. The scale bar is 10  $\mu$ m.

tetracyanoquinodimethane  $(F_4$ -TCNQ),<sup>16,17</sup> CF<sub>3</sub>-terminated thiol,<sup>18</sup> AuCl<sub>3</sub>,<sup>19</sup> and octadecyltrichlorosilane<sup>20,21</sup> for p-type dopants and nicotinamide adenine dinucleotide,<sup>16</sup> MgO,<sup>22</sup> aminopropyltriethoxysilane,<sup>21</sup> NH<sub>2</sub>-terminated thiol,<sup>18</sup> LiF,<sup>23</sup> hydrazine,<sup>24</sup> p-toluenesulfonic acid,<sup>25</sup> chloride molecules,<sup>26</sup> potassium,<sup>27</sup> and benzyl viologen (BV)<sup>28</sup> for n-type dopants. Among these candidates, BV has been widely proposed as a promising chemical doping material for n-type TMDs because it acts as an electron donor due to the difference in energy level between the reduction potential and the vicinity of the conduction band edge of TMDs.<sup>28,29</sup> Moreover, its excellent environmental stability provides a high degree of freedom from the perspective of manufacturing. For performing SCTD, organic dopants are typically deposited onto the surface of TMDs via simple solution immersion or dopant vaporization for a long time.<sup>20,21,24,27,28</sup> However, it is difficult to conduct chemical doping with high spatial selectivity on demand in these approaches, which is of fundamental importance to implement integrated circuit applications. Many studies based on conventional lithography have been carried out to enable selective chemical doping of TMDs. Laterally created 2D heterojunctions have been demonstrated by employing selective plasma treatments onto a MoS<sub>2</sub> flake with a spatially covered photoresist (PR) layer.<sup>30</sup> Additionally, selective chemical doping has been delivered with partially covered hexagonal boron nitride  $(h-BN)^{31}$  or via co-evaporation systems with a partially covered mask on TMDs.<sup>29</sup> Despite these efforts, new methodologies for introducing larger-area,

spatially selective SCTD on TMDs without the need for undesirable masks, complex patterning, or vacuum processing are highly required.

Additive manufacturing, especially in inkjet printing, can be a potential candidate due to its high-resolution, maskless, and drop-on-demand patterning ability with ink droplets having a volume of a few pL; therefore, precisely controlled selective doping can be performed on TMDs. There have been results of chemical doping *via* inkjet printing on oxide, organic, and carbon nanotube (CNT) thin-film transistors (TFTs) and their logic applications.<sup>32–34</sup> However, to the best of our knowledge, none have been reported for SCTD with inkjetprinted organic dopants on TMDs, even though efficient chemical doping is expected due to their high surface-tovolume ratio and atomically thin nature.

Herein, we demonstrate exquisitely controlled SCTD with BV inks for enhancing the electrical performance of CVDgrown MoS<sub>2</sub> FETs using spatially selective inkjet printing. The fluid characteristics of the BV ink are carefully optimized, which allows the creation of pL-scale ink droplets without satellites and results in a well-defined doping area on the MoS<sub>2</sub> channels. By exploiting the attractive merits of BV, inkjetprinted SCTD provides competitive doping effects to MoS<sub>2</sub> FETs, exhibiting efficient modulation of their electrical characteristics when compared with conventional chemical doping methods based on solution immersion and vaporization. Most importantly, drop-on-demand inkjet printing can directly deliver spatially controlled SCTD effects by selectively



Figure 2. (a) Schematic of the SCTD process on a  $MoS_2$  FET using inkjet printing. (b) Stable inkjettability window of the BV inks based on the capillary number (*Ca*)–Weber number (*We*) relationship of the toluene and cyclohexylbenzene mixtures. Time evolution of an ink droplet in flight after ejection from a nozzle for (c) BV ink based on toluene and (d) BV inks based on the toluene and cyclohexylbenzene mixture at a 4:6 ratio. The concentration of both BV inks is 5 mg/mL. An image of a single BV ink droplet on the  $MoS_2$  film by inkjet printing is also included in the inset. The scale bar is 20  $\mu$ m.

depositing BV dopants on  $MoS_2$  channels at different dopant concentrations. The changes in the electrical characteristics of  $MoS_2$  FETs, the shifts in Raman spectroscopy peak positions, the decrease in photoluminescence (PL) intensity, and the blue shift of the XPS binding energy of BV-doped  $MoS_2$  all support that controlling inkjet-printed SCTD depends on the BV concentration and the areal coverage of BV dopants. Our results demonstrate high potential for constructing 2D homojunctions in TMD films or complementary logic devices through the integration of SCTD-assisted 2D TMD-based FETs.

# **RESULTS AND DISCUSSION**

Figure 1a and b show the schematic of a MoS<sub>2</sub> FET with inkjet-printed Ag source/drain (S/D) electrodes and the optical image of the FET arrays. CVD-grown monolayer MoS<sub>2</sub> films were transferred onto 300-nm-thick SiO<sub>2</sub>/heavily doped Si substrates using the poly(methyl methacrylate) (PMMA)mediated transfer method.<sup>35</sup> For the formation of S/D contacts, a nanoparticle-type Ag ink was directly printed onto the MoS<sub>2</sub> channels without any surface treatment. Note that inkjet printing does not require procedures that are unfavorable to TMDs, such as residual chemical deposition and ultraviolet (UV) exposure.<sup>36</sup> Further details are described in the Methods section. The histogram of the field-effect mobility of the fabricated CVD-grown MoS<sub>2</sub> FETs with the inkjet-printed Ag electrode is shown in Figure 1c; notably, they are comparable to the field-effect mobility of conventional CVD-grown MoS<sub>2</sub> FETs fabricated by standard lithography.<sup>37</sup> Figure 1d and e show the representative transfer (sourcedrain current versus gate voltage,  $I_{DS}-V_{GS}$ ) and output (drainsource current versus drain voltage,  $I_{DS}-V_{DS}$ ) characteristics of the CVD-grown MoS<sub>2</sub> FETs with inkjet-printed Ag contacts. All electrical characterization was conducted in a vacuum at room temperature. From the transfer curve in Figure 1d, the field-effect mobility ( $\mu$ ) and 2D carrier density ( $n_{2D}$ ) can be calculated and are defined as<sup>28,38</sup>

$$\mu = \left(\frac{\partial I_{\rm DS}}{\partial V_{\rm GS}}\right) \frac{L}{W} \frac{1}{C_{\rm OX} V_{\rm DS}} \tag{1}$$

$$n_{\rm 2D} = I_{\rm DS} \frac{L}{W} \frac{1}{q \mu V_{\rm DS}} \tag{2}$$

where  $\partial I_{\rm DS}/\partial I_{\rm GS}$ , W, L, C<sub>OX</sub>, and q denote the transconductance, channel width, channel length, gate oxide capacitance, and electron charge, respectively. The extracted  $\mu$ , on/off ratio, subthreshold swing (SS), threshold voltage  $(V_{\rm th})$ , and  $n_{\rm 2D}$  are ~7.64 cm<sup>2</sup>/V·s at  $V_{\rm DS}$  = 60 V, 10<sup>7</sup>, 1.38 V/ decade, -13.1 V, and 2.6 × 10<sup>12</sup> cm<sup>-2</sup>, respectively. Specifically, in the logarithmic scale plot for  $I_{\rm DS}-V_{\rm DS}$  measured at  $V_{\rm GS}$  ranging from 0 to 2 V, the extracted average linearity parameter  $\gamma \sim 1$  indicates nearly ohmic contacts between the transferred MoS<sub>2</sub> channel and printed Ag contacts (Figure 1f). The quality of the transferred MoS<sub>2</sub> films was also evaluated by Raman and PL spectroscopy analysis (Figure 1g and h). The difference in wavenumber (~20  $\text{ cm}^{-1}$ ) between the in-plane  ${E_{2g}}^{1}$  and out-of-plane  $A_{1g}$  Raman peaks, the distinct PL peak at  $\sim$ 670 nm (corresponding to  $\sim$ 1.84 eV), and the PL mapping results of the channel region support that the transferred MoS<sub>2</sub> film is a uniformly grown monolayer (Figure 1i).<sup>39-43</sup>

Typically, BV molecules were dissolved into deionized water and became  $BV^{2+}$ ; then, nonpolar solvents and sodium borohydride were sequentially added into the BV solution as



Figure 3.  $I_{DS}-V_{GS}$  and  $I_{DS}-V_{DS}$  of MoS<sub>2</sub> FETs with and without SCTD conducted by (a, d) immersion in BV inks based on toluene for 3 h, (b, e) immersion in BV inks based on the toluene and cyclohexylbenzene mixture at a 4:6 ratio for 3 h, and (c, f) inkjet printing with BV inks based on the toluene and cyclohexylbenzene mixture at a 4:6 ratio. The concentrations of the BV inks are 5 and 20 mg/mL for immersion and inkjet printing, respectively.

a catalyst. After the redox reactions,  $BV^{2+}$  is reduced to hydrophobic  $BV^0$ , allowing its separate transfer from water to nonpolar solvents. More details are described in the Methods section. The conduction band edge ( $E_C$ ) of MoS<sub>2</sub> is located at ~0 V versus the standard hydrogen electrode,<sup>44,45</sup> while the reduction potentials of the BV molecule throughout the two successive redox reactions from  $BV^+$  to  $BV^0$  and from  $BV^{2+}$  to  $BV^+$  are equal to -0.79 and -0.33 V, respectively (Figure S1 in the Supporting Information).<sup>46</sup> Due to the band offset where the reduction potential of the BV molecule is positioned below the  $E_C$  edge of MoS<sub>2</sub>, electrons can be transferred from the neutral  $BV^0$  to the MoS<sub>2</sub> channel, resulting in a significant increase in carrier concentration.

To selectively conduct SCTD with BV onto CVD-grown  $MoS_2$  channels *via* inkjet printing as shown in Figure 2a, rheologically apposite BV inks need to be prepared to achieve high-quality inkjet printing that is closely related to the resolution of the doping area. One of the most critical factors is to create stable ink droplets having volumes of a few pL with excellent inkjetability, which can be estimated by the Ohnesorge number (*Oh*). *Oh* can be calculated from three dimensionless parameters, namely, the Reynolds number (*Re*), Weber number (*We*), and capillary number (*Ca*), which are determined by the fluidic properties of inks, including their viscosity, surface tension, density, and inertial force.

$$Re = \frac{\text{inertial force}}{\text{viscous force}} = \frac{\rho v d}{\eta}$$
(3)

$$We = \frac{\text{inertial force}}{\text{surface tension force}} = \frac{\rho v^2 d}{\gamma}$$
(4)

$$Ca = \frac{\text{viscous force}}{\text{surface tension force}} = \frac{\eta \nu}{\gamma}$$
(5)

$$Oh = \frac{\text{viscous force}}{\sqrt{\text{inertial force } \times \text{ surface tension}}} = \frac{\eta}{\sqrt{\sigma\rho d}}$$
(6)

where  $\rho$  is the density of the ink,  $\nu$  is the velocity of the ink, d is the nozzle diameter,  $\eta$  is the viscosity of the ink, and  $\gamma$  is the surface tension of the ink. It is well known that the inverse of Oh(Z) ranging from 1 to 10 can be a criterion for reliable inkjettability without undesirable satellite droplets.<sup>47</sup> Although toluene has been widely used to prepare a BV ink as a nonpolar solvent, it is unfavorable to form well-defined inkjet droplets with a Z value of  $\sim 27$ , resulting in unstable inkjetting, as shown in Figure 2c. To address this issue, we modified the solvent system of BV ink while simultaneously considering BV solubility and inkjettability. Cyclohexylbenzene (CHB) can be a promising candidate as an additive due to its high compatibility with toluene, apposite Z value of 6.45, and high boiling point of 238 °C, all of which lead to reliable inkjetting.<sup>50,51</sup> Furthermore, the CHB-toluene mixed solvent (simply denoted as the mixed solvent) promotes more uniformly deposited BV layers by competing for the Marangoni effect and capillary action (Figure S3 in the Supporting Information). Therefore, BV ink was dissolved in the mixed solvent to meet the requirements mentioned above (Figure 2b). From the theoretical design and experimental results, a 4:6 volume ratio of toluene to CHB is determined with a Z value of 8.98, as shown in Table S1 in the Supporting Information. Our solvent system produces well-defined and stable BV ink droplets with a volume of  $\sim 2 \text{ pL}$  (Figure 2d), and these droplets form a circular pattern with a diameter of ~40  $\mu$ m on the MoS<sub>2</sub> channel.

To investigate the compatibility of the CHB-toluene mixture as a solvent of BV ink and inkjet printing to deliver the n-type doping effects, SCTD was conducted on CVD-grown MoS<sub>2</sub> FETs *via* three methods: immersion in a BV ink based on toluene, immersion in a BV ink based on the mixed solvent, and inkjet printing of BV ink based on the mixed solvent. These methods were followed by electrical characterization in a vacuum. Figure 3 shows the electrical characteristics of the MoS<sub>2</sub> FETs before and after SCTD, including the transfer curves (Figure 3a-c) and output characteristics (Figure 3d-f) for three methods. All three SCTD processes deliver consistent n-type doping effects to the MoS<sub>2</sub> FETs,



Figure 4. (a) Optical image of spatially selective BV-doped  $MoS_2$  obtained by inkjet printing. (b) PL mapping image showing the peak intensity at a photon energy of 1.84 eV (A exciton). Raman mapping image of the Raman shift of the (c)  $E_{2g}^{-1}$  and (d)  $A_{1g}$  modes. (e) PL spectra of the pristine and BV-doped  $MoS_2$  films. (f) Raman spectra of the pristine and BV-doped  $MoS_2$  films. (f) Raman spectra of the pristine and BV-doped  $MoS_2$  films. The symbols are experimental data, and the solid lines are the fitting results with a Gaussian function. The scan size of PL and Raman mapping is 100  $\mu$ m × 200  $\mu$ m. The concentration of BV inks is 5 mg/mL for SCTD on  $MoS_2$ . The scale bar is 100  $\mu$ m.



Figure 5. Deconvoluted XPS spectra of the pristine and BV-doped  $MoS_2$  films for the (a) N 1s peaks, (b) Mo 3d and S 2s peaks, and (c) S 2p peaks. The symbols and solid lines are the experimental XPS data and deconvolution of the XPS spectra using the Gaussian function, respectively.

exhibiting a  $V_{\rm th}$  shift toward the negative gate voltage direction and a significant increase in the field-effect mobility, on-state current, and carrier sheet density (Table S2 in the Supporting Information for details). These results support that the prepared BV inks for stable inkjet printing can be successfully introduced to modulate the electrical performance of MoS<sub>2</sub> FETs *via* spatially selective SCTD. The increase of electron mobility after doping could be attributed to the reduction of contact resistance and the screening effect of excess doped electrons.<sup>19,23,26,28</sup> Further improvements would be expected if remote modulation doping can be introduced onto 2D materials by suppressing charge impurity scattering in the channel.<sup>52</sup> It should also be noted that no significant change is observed in the electrical characteristics of  $MoS_2$  FETs after immersion in the solvent only (without BV molecules) (Figure S4 in the Supporting Information).

The changes in the optoelectronic properties of  $MoS_2$  after inkjet printing BV molecules were also investigated using Raman and PL spectroscopy. The BV ink was inkjet-printed with high spatial selectivity on the  $MoS_2$  channel, as illustrated in Figure 4a. Figure 4b–d show the PL mapping and Raman mapping images of a specific region of  $MoS_2$  film with and without inkjet-printed BV molecules indicated as a dashed box in Figure 4a. As shown in Figure 4b and e, the A exciton peak



Figure 6. (a) Optical images of the BV inks with different BV concentrations. (b) Spatially selected SCTD with different areal coverages *via* inkjet printing. (c) Transfer curve of the BV-doped MoS<sub>2</sub> FET with different BV concentrations *via* inkjet printing. The areal coverage ratio is 1.00. (d) Transfer curve of the BV-doped MoS<sub>2</sub> FET with different areal coverages *via* inkjet printing. The concentration of BV ink is 5 mg/mL. (e) Optical image of the BV-doped MoS<sub>2</sub> FET array with different BV concentrations and areal coverage *via* spatially selective inkjet printing. The contour plot indicates the ratio of the on-state current of 80 BV-doped MoS<sub>2</sub> FETs as a function of the doping concentration and areal coverage at a fixed  $V_{DS}$  of 1 V and  $V_{GS}$  of 60 V. The scale bar is 200  $\mu$ m.

of ~1.84 eV corresponding to the optical bandgap of monolayer  $MoS_2$  is observed throughout the undoped area of  $MoS_2$ , while the BV-doped  $MoS_2$  uniformly exhibits a decreased PL intensity with a red shift in the A exciton energy. This decrease in the A exciton arises from the suppression of radiative recombination by excess electrons due to chemical doping, consistent with the results in previous studies.<sup>16,53,54</sup> Likewise, the results of Raman spectroscopy also support the effect of inkjet-printed SCTD with BV. It is known that the out-of-plane  $A_{1g}$  is more sensitive to the presence of deposited molecular dopants on the surface of  $MoS_2$  than the in-plane  $E_{2g}^{-1}$  due to stronger electron—phonon coupling.<sup>42,43</sup> We also observe that the  $A_{1g}$  peak shifts to the lower wavenumber direction, while the position of the  $E_{2g}^{-1}$  peak is unchanged after performing inkjet-printed SCTD, as shown in Figure 4f.

Furthermore, X-ray photoelectron spectroscopy (XPS) analysis reveals that the inkjet-printed BV dopant donates electrons to the underlying  $MoS_2$  film. Figure 5a-c show the XPS spectra of the undoped  $MoS_2$  (top) and BV-doped  $MoS_2$  (bottom) films for Mo, S, and N, respectively. The XPS spectra show two distinctive photoelectron signals at binding energies of 399.8 and 402.0 eV related to N 1s after introducing the inkjet-printed BV; notably, these peaks are not observed on undoped  $MoS_2$ . Thus, these two peaks are attributed to viologen nitrogen radicals (BV<sup>+</sup>) and positively charged viologen nitrogen radicals (BV<sup>2+</sup>), which support electron donation from the inkjet-printed BV dopants.<sup>55</sup> In addition, the representative XPS peaks of  $MoS_2$  shift toward a higher binding energy with inkjet-printed SCTD (Figure 5b and c),

which indicates that the Fermi level of the  $MoS_2$  energy band moves to the conduction band edge due to the transferred charges from the BV dopants to  $MoS_2$ .<sup>56</sup>

With the advantages of the BV inks and inkjet printing, we demonstrate elaborate control of the SCTD effect on the electrical performance of MoS<sub>2</sub> FETs by adjusting the dopant concentration and areal coverage of BV dopants. Figure 6a shows the BV inks dissolved in the mixed solvent with different dopant concentrations: 1, 5, 10, and 20 mg/mL. Because the CVD-grown MoS<sub>2</sub> channels have a high surface-to-volume ratio with a 2D nature, the SCTD effects are highly responsive to the areal coverage of BV dopants as well as the BV concentration. In this regard, the electrical characteristics of BV-doped MoS<sub>2</sub> FETs can be modulated by changing the deposited BV dopants with an areal coverage ratio of 0.25 to 1.00. This is feasible due to the high spatial resolution of welloptimized inkjet printing, which can achieve a scale of tens of  $\mu$ m on demand, as shown in Figure 6b. Figure 6c shows the transfer characteristics of pristine MoS<sub>2</sub> FETs and BV-doped MoS<sub>2</sub> FETs with different BV concentrations of 1, 5, 10, and 20 mg/mL by inkjet printing. As the dopant concentration increases, stronger n-type doping effects, such as a shift in  $V_{\rm th}$ toward the negative gate voltage direction, are observed, which indicates a higher carrier concentration and an increase in the on-state current. Figure 6d shows the representative transfer curves of pristine MoS<sub>2</sub> FETs and BV-doped MoS<sub>2</sub> FETs with different areal coverages of BV dopants with a fixed BV concentration of 5 mg/mL. The controlled n-type doping effects depending on the BV concentration and the areal

coverages of BV dopants are summarized in Table S3 and Figure S5 in the Supporting Information. In previous studies, although many efforts have been dedicated to optimizing a doping process with different immersion times and concentrations of BV dopants, there are inherent limitations: it is difficult to deliver exquisitely controlled doping effects to  $MoS_2$ FET arrays with high spatial selectivity on a single substrate. In this work, by adjusting the amount of dopants physically adsorbed to the  $MoS_2$  channel *via* inkjet printing, we demonstrate an array of CVD-grown  $MoS_2$  FETs selectively doped with different SCTD conditions on a single substrate. The contour plot indicates the gradually tailored on-state current of BV-doped  $MoS_2$  FETs as a function of the doping concentration and areal coverage (Figure 6e).

# CONCLUSIONS

In conclusion, we introduced controllable SCTD for tailoring the electrical characteristics of MoS<sub>2</sub> FETs using spatially selective inkjet printing. The solvent system to prepare inkjetprintable BV inks was optimized considering their fluid properties, which allowed high spatial selectivity and excellent inkjettability with well-defined droplets having a volume of a few pL. Inkjet-printed SCTD with BV delivered efficient n-type doping effects to MoS<sub>2</sub> FETs, achieving results that were comparable to those of SCTD conducted by using conventional solution immersion for a long time. Drop-on-demand inkjet printing and the high degree of freedom of BV concentrations enabled us to elaborately control SCTD to produce MoS<sub>2</sub> FETs with well-tailored electrical characteristics, which cannot be achieved via conventional chemical doping based on solution immersion and vaporization. The Raman and PL spectroscopy results could be evidence to support the SCTD effect of inkjet-printed BV dopants. We believe the suggested inkjet-printed SCTD method can be applied to other organic dopants if their solubility and stability in inkjet-printable solvents are guaranteed. Our study proposes a feasible solution for controllable SCTD using inkjet printing toward the realization of further practical TMD-based devices with efficiently tailored electrical characteristics.

# **METHODS**

Procedure of MoS<sub>2</sub> FET Fabrication. Poly(methyl methacrylate) was spin-coated as a sacrificial layer onto a continuous CVDgrown monolayer MoS<sub>2</sub> film on a SiO<sub>2</sub>/heavily doped silicon substrate (purchased from Six-Carbon Corp.), followed by hard baking on a 180 °C hot plate for 90 s. Then, thermal releasing supporting tape (purchased from Graphene Square) was directly attached to the PMMA-coated CVD-grown MoS2 film. It was immersed in potassium hydroxide solution (KOH) at 60 °C to remove the SiO<sub>2</sub> layer, rinsed with deionized water, and dried with a  $N_2$  gun. The PMMA-coated  $MoS_2$  film was then transferred onto the target substrate and dried in a vacuum oven for 12 h. After removing the supporting tape through thermal treatment on a 120 °C hot plate, finally, the PMMA sacrificial layer was immersed in 60 °C acetone for 10 min to reproduce monolayer MoS<sub>2</sub> films. The electrical contacts were inkjet-printed directly on the transferred MoS<sub>2</sub> with nanoparticle-type Ag ink (DGP 40LT-15C, ANP Co. Ltd.). The silver ink was printed with a drop velocity of ~3 m/s, drop frequency of 3 kHz, and drop spacing of 25  $\mu$ m (1016 dpi) under a printing bed at 60 °C. After the source and drain electrodes were printed, they were annealed at 150  $^\circ \mathrm{C}$  for 30 min on a hot plate. The electrical characteristics of the pristine MoS<sub>2</sub> FETs and doped MoS<sub>2</sub> FETs were measured using a semiconductor parameter analyzer (Agilent B1500A) under vacuum conditions.

**Preparation of Benzyl Viologen Inks.** Benzyl viologen dichloride powders were dissolved in 5 mL of deionized water followed by the addition of 5 mL of nonpolar solvent (pure toluene, the mixed solvent: toluene and cyclohexylbenzene at a 4:6 ratio) for a biphasic solution. Sodium borohydride powder (Sigma–Aldrich), which acted as a catalyst, was added to the nonpolar solvent–water biphasic solution. After storing for 1 day, the top nonpolar solution containing the BV dopant was extracted using a syringe. The chemical doping of  $MoS_2$  FETs was conducted either by immersion into the BV solution for 3 h or inkjet printing of the BV ink. Then, BV-doped  $MoS_2$  FETs were annealed at 120 °C for 1 h in a vacuum oven to eliminate residual solvents. All materials were purchased from Sigma–Aldrich Corp.

**Characterization of the Pristine or BV-Doped MoS**<sub>2</sub>. Raman and PL spectroscopy were performed to investigate the optoelectronic properties of the MoS<sub>2</sub> film. Raman spectroscopy was performed using an XperRam 200 instrument (Nanobase, Inc.) with a laser source excitation wavelength of 532 nm under ambient conditions at room temperature. A 40× objective lens was used to focus the laser at a particular spot on the MoS<sub>2</sub> film (~1  $\mu$ m spot radius). XPS of the MoS<sub>2</sub> films was performed with a Thermo Fisher Scientific X-ray photoelectron spectrometer system utilizing a monochromatic Al K-alpha radiation source. A pass energy of 50 eV with a 0.1 eV scanning step was used for photoelectron detection. XPS spectra were obtained with a beam spot size of 50 × 50  $\mu$ m<sup>2</sup>. All XPS spectra were calibrated using the carbon (C) 1s peak at a binding energy of 284.4 eV as a reference.

# ASSOCIATED CONTENT

#### **Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acsnano.2c00021.

Details on surface charge transfer doping with the BV dopants, diagram of pulse waveform for reliable inkjet printing, BV film formation depending on the ink solvent, ink jettability depending on the volume ratio of toluene to cyclohexylbenzene, electrical characteristics of the pristine  $MoS_2$  FETs and BV-doped  $MoS_2$  FETs by solution immersion and inkjet printing, electrical characteristics of  $MoS_2$  FETs after immersing into pure toluene and the CHB–toluene mixed solvent, and electrical characteristics of the  $MoS_2$  FETs before and after spatially controllable BV doping by inkjet printing (PDF)

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# Notes

The authors declare no competing financial interest.

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