

Molecular Dopant-Dependent Charge Transport in Surface-Charge-Transfer-Doped Tungsten Diselenide Field Effect Transistors

Jae-Keun Kim, Kyungjune Cho, Juntae Jang, Kyeong-Yoon Baek, Jehyun Kim, Junseok Seo, Minwoo Song, Jiwon Shin, Jaeyoung Kim, Stuart S. P. Parkin, Jung-Hoon Lee,* Keehoon Kang,* and Takhee Lee*

The controllability of carrier density and major carrier type of transition metal dichalcogenides(TMDCs) is critical for electronic and optoelectronic device applications. To utilize doping in TMDC devices, it is important to understand the role of dopants in charge transport properties of TMDCs. Here, the effects of molecular doping on the charge transport properties of tungsten diselenide (WSe₂) are investigated using three p-type molecular dopants, 2,3,5,6-tetrafluoro-7,7,8,8-tetracyanoquinodimethane (F₄-TCNQ), tris(4-bromophenyl) ammoniumyl hexachloroantimonate (magic blue), and molybdenum tris(1,2bis(trifluoromethyl)ethane-1,2-dithiolene) (Mo(tfd-COCF₃)₃). The temperaturedependent transport measurements show that the dopant counterions on WSe₂ surface can induce Coulomb scattering in WSe₂ channel and the degree of scattering is significantly dependent on the dopant. Furthermore, the quantitative analysis revealed that the amount of charge transfer between WSe2 and dopants is related to not only doping density, but also the contribution of each dopant ion toward Coulomb scattering. The first-principles density functional theory calculations show that the amount of charge transfer is mainly determined by intrinsic properties of the dopant molecules such as relative frontier orbital positions and their spin configurations. The authors' systematic investigation of the charge transport of doped TMDCs will be directly relevant for pursuing molecular routes for efficient and controllable doping in TMDC nanoelectronic devices.

1. Introduction

2D layered materials have attracted remarkable attention due to their ultrathin nature with thickness-dependent tunable electronic structure which make them a promising candidate for the next-generation electronic and optoelectronicdevices.^[1–13] In particular, semiconducting transition metal dichalcogenides (TMDCs), such as molybdenum disulfide (MoS₂) and tungsten diselenide (WSe₂), have shown desirable field-effect transistor (FET) properties such as the high on–off ratio, high mobility, and low subthreshold swing voltage.

To utilize TMDCs for various electronic and optoelectronic device applications, the ability to control conductivity and carrier type is essential. In this manner, several studies have been reported for controlling the electrical properties of TMDCs by doping which allows tuning of free carrier concentration and the majority carrier type.^[14–23] One of the approaches is substitutional doping, which is replacing transition metal elements (Mo, We) in

J.-K. Kim, J. Jang, K.-Y. Baek, J. Kim, J. Seo, M. Song, J. Shin, J. Kim, T. Lee

Department of Physics and Astronomy and Institute of Applied Physics Seoul National University Seoul 08826, Korea

E-mail: tlee@snu.ac.kr

J.-K. Kim, K. Cho, S. S. P. Parkin

Max-Planck Institute of Microstructure Physics Weinberg 2, 06120 Halle (Saale), Germany

K. Cho

Soft Hybrid Materials Research Center Korea Institute of Science and Technology Seoul 02792, Korea

The ORCID identification number(s) for the author(s) of this article can be found under https://doi.org/10.1002/adma.202101598.



I.-H. Lee

K. Kang

Seoul 02792, Korea

Yonsei University

Seoul 03722, Korea

E-mail: jhlee84@kist.re.kr

Computational Science Research Center

E-mail: keehoon.kang@yonsei.ac.kr

Korea Institute of Science and Technology

Department of Materials Science & Engineering



TMDCs with other elements with a different number of valence electrons (such as Nb, Re, or Zr).^[14-16] Although substitutional doping has an advantage in terms of the doping stability, the controllability of doping level is comparatively difficult to achieve via a direct synthesis.^[15,24] Another route for controlling the electrical characteristics of TMDCs is by surface charge transfer doping (SCTD).^[17-23] SCTD involves physisorption of dopants on the surface of TMDCs, which induces a charge transfer across the TMDC/dopant interface. A large surface area to volume ratio of 2D layered TMDCs makes them highly sensitive to the surface-attached dopants. In addition, comparing to substitutional doping, SCTD has an advantage that it is a relatively non-destructive process. Particularly, for SCTD by molecular reductants (oxidants) for n(p)-type doping, the potential range of material selection for dopants is diverse due to a high degree of freedom in the molecular design.^[25] For these reasons, several studies have demonstrated the applicability of molecular SCTD in TMDC electronic devices for reducing contact resistance and controlling the carrier type for complementary logic applications.^[19-21,26-28]

Considering that SCTD involves the creation of charged dopants on the surface of the doped TMDCs, it is natural to consider the Coulomb scattering induced by the dopants in the proximity to the charge carriers in TMDC. In this regard, in graphene, SCTD was shown to induce charged impurity scattering which significantly affected the charge transport.^[29] Likewise, we can expect that SCTD will induce charged impurity scattering in doped TMDCs. Ideally, molecular SCTD should preserve the high carrier mobility of TMDCs by minimizing the charged impurity scattering.^[30,31] Therefore, in order to fully realize the potential of molecular doping, it is essential to understand what fundamental properties of molecular dopants and their interfacial properties with TMDCs determine the degree



of charge transfer, as well as their effects on the overall charge transport, both of which have been rarely studied. In particular, a direct relationship between local charge transfer processes of different molecular dopants and their effects on the charge transport properties of TMDC devices is yet to be quantified. Here, we systematically studied the electrical transport in doped WSe₂ by SCTD using three p-type molecular dopants with different electron affinities. Considering that doping strength of each molecular dopant is typically dependent on intrinsic molecular properties such as their relative frontier orbital positions, we investigated the doping effects on the charge transport of WSe₂ FETs by measuring the temperature-dependent intrinsic channel mobility values. Additionally, we performed first-principle density functional theory (DFT) calculations to compute and compare the degrees of charge transfer from WSe2 to molecular dopants, which could be related to a varying degree of charged-impurity scattering induced by each dopant ion.

2. Results and Discussion

As the p-type molecular dopants for SCTD on WSe₂, three different molecules were used in this study; 2,3,5,6-tetra-fluoro-7,7,8,8-tetracyanoquinodimethane (F_4 -TCNQ), tris(4-bromophenyl)ammoniumyl hexachloroantimonate (magic blue), molybdenum tris(1,2-bis(trifluoromethyl)ethane-1,2-dith-iolene) (Mo(tfd-COCF₃)₃). The molecular structure of each dopant is shown in **Figure 1**a. The dopants with different lowest unoccupied molecular orbital (LUMO) levels were chosen for investigating different doping strengths according to the LUMO level which correlates with the electron affinity. Figure 1b shows the energy diagrams for the valence band maximum (VBM) of WSe₂ and LUMO levels of the molecular



Figure 1. a) The lattice structure of WSe₂ and the molecular structures of F_4 -TCNQ, magic blue, and Mo(tfd-COCF₃)₃ molecular dopants. b) Energy diagram of WSe₂ and the LUMO levels of dopant molecules. CBM and VBM are the abbreviations of conduction band minimum and valence band maximum, respectively. c) The schematic of four-point probe measurement with optical image of a WSe₂ FET.

www.advancedsciencenews.com





Figure 2. The representative transfer curves of pristine and p-doped WSe₂ FETs by a) F4-TCNQ, b) magic blue, and c) Mo(tfd-COCF₃)₃ with three different dopant solution concentration. d) The carrier density generated by F₄-TCNQ, magic blue, and Mo(tfd-COCF₃)₃ doping (Δn_{WSe_2}) with three different dopant solution concentration. Error bars indicate the standard deviation from the mean value extracted from several devices.

dopants. Due to similar (F₄-TCNQ) or deeper (magic blue and Mo(tfd-COCF₃)₃) LUMO levels of the dopants relative to the VBM of WSe₂ (5.33 eV),^[32,33] all three molecules are expected to act as p-type dopants. Also, it is expected that magic blue (LUMO level: 5.5 eV)^[21] and Mo(tfd-COCF₃)₃ (5.6 eV)^[34] generate a higher carrier density in WSe₂ FETs than F₄-TCNQ (5.2 eV).^[35] To measure the intrinsic channel mobility of WSe₂ FETs, we employed a four-point probe measurement as schematically shown in the optical image of a WSe₂ FET device in Figure 1c. In this measurement, the intrinsic conductance of WSe₂ channel (i.e., free from contact resistance) can be obtained by measuring the voltage difference between the two inner probes (P_1, P_2) located in the channel while current was flown between the contacts (S, D) by applying drain (V_D) and gate voltages (V_{GS}) . In device fabrication, we confirmed that the effect of thickness of WSe2 on doping was not dominant in our devices (see details in Experimental Section and Supporting Information).

Before we conducted the four-point probe measurement, the effects of the molecular dopants were investigated through conventional FET measurement with two-probe measurements. **Figure 2**a–c shows representative transfer curves of pristine WSe₂ FETs and p-doped WSe₂ FETs with F₄-TCNQ, magic blue, and Mo(tfd-COCF₃)₃ at different doping levels. The pristine WSe₂ FETs exhibited typical ambipolar characteristics.^[36,37] As shown in Figure 2a, for F₄-TCNQ molecular solution the threshold voltage shifts to the positive gate voltage direction, indicating the typical p-type doping behavior. As the dopant solution concentration of F₄-TCNQ increases from 1 to 5 mM, the current level of the WSe₂ FET increases and the threshold voltage shifts to the higher positive gate voltage. This

result demonstrates the controllability of F_4 -TCNQ doping via a sequential doping with the varying solution concentration. The same doping processes were performed for the other two molecular dopants (magic blue and Mo(tfd-COCF₃)₃) in the same manner as F_4 -TCNQ. As shown in Figure 2b,c, the overall results exhibit similar characteristics for the other two dopants; the current level increases and the threshold voltage shifts to positive gate voltage direction with increasing dopant solution concentration. We can clearly see consistent results for several devices (see Figure S2, Supporting Information). Furthermore, we confirmed the p-type doping behavior by the shift of work function of WSe₂ after doping using the Kelvin probe force microscope measurement (see Figure S3, Supporting Information).

To compare the doping strength and doping range achievable with each molecular dopant, we have calculated the amount of generated carrier density by doping (Δn_{WSe2}) using the following relation $\Delta n_{WSe2} = C_i \Delta V_{th}$, where C_i is the capacitance per unit area of the bottom SiO₂ layer ($C_i = [\varepsilon_0 \ \varepsilon_r]/d =$ 1.3×10^{-4} F m⁻², where $\varepsilon_{\rm o}$ is the permittivity of vacuum $[8.85 \times 10^{-12} \text{ F m}^{-1}]$, ε_r is dielectric constant of SiO₂ [3.9], and d is thickness of SiO₂ [270 nm]) and ΔV_{th} is the shift in the threshold voltage of the doped WSe₂ FETs relative to the pristine devices. $\Delta n_{\rm WSe2}$ for three molecular dopants cases are summarized in Figure 2d. For this figure, several WSe₂ FETs (typically four to six devices) were measured to obtain the average $\Delta V_{\rm th}$ and standard deviation values shown as error bars. For all the three molecular dopants, $\Delta n_{\rm WSe2}$ increases with increasing the concentration of the dopant solutions, as shown in Figure 2d. It is worth noting that the maximum $\Delta n_{\rm WSe2}$ for F₄-TCNQ (3.61 × 10¹² cm⁻²) was lower than that







Figure 3. The representative gate-voltage dependent conductance curves of a) pristine and b) $Mo(tfd-COCF_3)_3$ -doped WSe₂ devices at various temperatures. The representative temperature-dependent intrinsic mobility values extracted for the pristine and doped WSe₂ devices with c) F₄-TCNQ, d) magic blue, and e) $Mo(tfd-COCF_3)_3$.

for magic blue $(7.03 \times 10^{12} \text{ cm}^{-2})$ and Mo(tfd-COCF₃)₃ devices $(4.15 \times 10^{12} \text{ cm}^{-2})$. Interestingly, although the LUMO levels of magic blue and Mo(tfd-COCF₃)₃ are comparable, the maximum Δn_{WSe2} for magic blue is about 1.7 times larger than that for Mo(tfd-COCF₃)₃. This indicates that the relative LUMO level of the dopant molecules with respect to the VBM of WSe₂ is not the sole parameter that determines the doping strength. In other words, we need to consider both extrinsic (e.g., coverage of the dopant molecules on the surface of TMDCs^[38,39]) and intrinsic factors (e.g., relative frontier orbital position) for doping. In this regards, we conducted X-ray photoelectron spectroscopy (XPS) and atomic force microscope (AFM) for investigating the surface coverage of dopant molecules (see Section S4, Supporting Information). In the surface coverage analysis with XPS and AFM, we confirmed that the dopants are not distributed uniformly on the WSe₂ surface but instead form grain-like structures. In addition, multilayer stacking of dopant molecules was indicated by the AFM measurements similarly to a previous report.^[40] The larger coverage of magic blue dopant molecules compared to that of Mo(tfd-COCF₃)₃ can explain the higher maximum $\Delta n_{\rm WSe2}$ when doped with magic blue dopants (see Figure S4, Supporting Information).

Because both the channel resistance and contact resistance are affected after doping (Figure S6, Supporting Information), the doping effect to the channel resistance should be separated from that to the contact resistance for investigating the effect of doping to charge transport of WSe₂. In this study, we focus on the intrinsic mobility changes of the channel by eliminating the contribution of the contact resistance toward the total resistance of the device (see Figure 1c). Using four-point probe measurement configuration, we conducted temperature-dependent electrical characterization. **Figure 3**a,b shows typical gate-voltage dependent conductance curves of pristine and Mo(tfd-COCF₃)₃-doped WSe₂ devices, respectively, at different temperatures. The temperature-dependent conductance curves of the doped WSe₂ with the other dopants (F₄-TCNQ and magic blue) are provided in the Figure S7 of Supporting Information. The channel mobility can be calculated from the conductance curve by $\mu_{FET} = (dG/dV_{GS}) \times (L/WC_i)$, where $C_i = 1.3 \times 10^{-4}$ F m⁻² and *G*, *L*, and *W* denote the conductance, the channel length, and the channel width, respectively. Figure 3c–e shows the temperature-dependent intrinsic mobility values for the pristine and doped WSe₂ devices with 2.5 mM solution. To investigate different scattering mechanisms in the doped WSe₂ devices, we decompose the temperature dependence of the mobility by using Matthienssen's rule

$$\mu_{FET}(T) = \left(1/\mu_{\rm C}T^{\alpha} + 1/\mu_{Ph}T^{\beta}\right)^{-1}$$
(1)

where *T*, $\mu_{\rm C}$, $\mu_{\rm Ph}$, and α and β denote temperature, charged impurity scattering-limited mobility at the zero temperature, phonon-limited mobility at the zero temperature, and their exponents, respectively. The contribution from each scattering mechanism (i.e., charged impurity scattering $[\mu_{\rm C}T^{\alpha}]$ and phonon scattering $[\mu_{\rm Ph}T^{\beta}]$) are shown in Figure 3c–e as dashed lines. The values of these components are summarized in Table S5 of Supporting Information. In this analysis, we assume that the scattering from intrinsic defects in WSe₂ (e.g., Se vacancies^[41]) and other scattering sources near the substrate were negligible in our multilayer WSe₂ FET devices.^[42] These are relevant since the charged impurity scattering (i.e., a long-range scattering) and phonon scattering







Figure 4. a) The carrier density dependent calculated WSe₂ mobility with different n_{imp} (lines) and the experimentally measured intrinsic mobility of pristine and doped WSe₂ (circles). b) The plot of n_{imp} generated in WSe₂ FETs by doping (Δn_{imp}) with F₄-TCNQ, magic blue, and Mo(tfd-COCF₃)₃ dopants versus generated carrier density by doping (Δn_{WSe_2}).

contributions are the most dominant mechanisms in the charge transport of TMDCs. $^{[30,41]}$

In the case of multilayer TMDCs, a homopolar phonon mode can be dominant among the phonon modes to determine the intrinsic mobility of TMDCs.^[28] Rai et al. reported that the homopolar phonon mode of MoS₂ can be suppressed by depositing a high-k amorphous titanium suboxide on top of the MoS₂ surface, which was confirmed by decreasing of $\beta^{[43]}$ However, β value barely changed upon doping in WSe₂ devices for all the dopants (see Figure 3c-e and Table S5, Supporting Information). This suggests that the phonon scattering contribution to the charge transport properties of the WSe₂ is almost insensitive to the presence of the dopant molecules. On the contrary, the charged impurity scatteringlimited mobilities of WSe₂ ($\mu_{\rm C}T^{\alpha}$ in Figure 3c–e) were generally reduced after doping for all three dopants (i.e., the charged impurity scattering increased upon doping). These results indicate that the dopant counterions created on the WSe₂ surface can act as charged impurity scattering centers for the WSe₂ channel. The mobility reduction by the externally introduced charged impurities has been well established for various semiconductor materials (e.g., Si and GaAs).^[44-46] We note that the magnitude of the reduction in the charged-impurity-scattering-limited mobility (i.e., $\mu_{\rm C} T^{\alpha}$) is different for each dopant, indicating that the degree of scattering in the WSe₂ channel varies with the dopant molecule.

To quantitatively analyze the degree of charged impurity scattering, we have extracted the charged impurity density present in the doped WSe₂ by relating to their transport properties. Ong et al. reported a theoretical model for calculating the charged impurity-limited mobility (μ_{imp}) of TMDCs.^[31] Using this model, we obtained the charged impurity of pristine and doped WSe₂ from the intrinsic mobility value of WSe₂ at 10 K where charged impurity scattering should be the most dominant in the transport with a negligible phonon scattering. In the model, the charged impurity-limited mobility of TMDCs is determined by the scattering rate for the charged impurity scattering rate for

the single charged impurity ($\Gamma_{\rm imp}$) can then be given by the following equation:

$$\Gamma_{\rm imp}(E_k) = \frac{1}{2\pi\hbar} \int d\mathbf{k}' \left| \phi_{\mathbf{k}-\mathbf{k}'}^{\rm scr} \right|^2 \times (1 - \cos\theta_{\mathbf{k}\mathbf{k}'}) \delta\left(E_{\mathbf{k}} - E_{\mathbf{k}'}\right)$$
(2)

where $\theta_{kk'}$ is the scattering angle between the k and k' states. From Γ_{imp} , the charged impurity-limited mobility (μ_{imp}) is given by

$$\mu_{\rm imp} = \frac{e}{\pi n_{\rm WSe_2} \hbar^2 k_{\rm B} T} \int_0^\infty f(E) [1 - f(E)] (n_{\rm imp-doped\,(pristine)} \Gamma_{\rm imp}(E))^{-1} E dE$$
(3)

where f(E) is the Fermi-Dirac function, $k_{\rm B}$ is the Boltzmann constant, n_{WSe2} is the carrier density, and $n_{imp-doped (pristine)}$ is the charged impurity density of doped (or pristine) channel. More details about the theoretical model are explained in the Supporting Information (Figure S8, Supporting Information). Figure 4a shows the calculated μ_{imp} from the model as a function of the carrier density in the WSe2 channel, $n_{\rm WSe2}$, for selected five different values of charged impurity density ($n_{\rm imp}$ = 1, 2, 3, 4, and 6 × 10¹² cm⁻²). To extract the experimentally determined charged impurity density for each device (N_{imp}) , we compared the measured mobility values (μ_{FET}) (circles) obtained from each device to the calculated μ_{imp} values (blue lines) shown in Figure 4a. Representative mobility and carrier density (n_{WSe2}) values of the pristine (doped) WSe₂ devices obtained at $V_{\rm G} = -120$ V are plotted as open (filled) circles in Figure 4a. A set of pristine and doped WSe₂ for each molecular dopant is denoted by different colors. As shown in Figure 4a, the mobility reduction upon doping for all three dopants, as indicated by the arrows, is a direct evidence of the Coulomb scattering induced by the dopant counterions in the doped WSe₂ devices. Therefore, we can estimate the extra impurity density generated by doping, $\Delta N_{\rm imp}$ (= $N_{\rm imp-doped}$ – $N_{\rm imp-pristine}$), for each dopant from the mobility curves for various n_{imp} values as shown in Figure 4a,







Figure 5. Computed planar-averaged charge density difference profiles along the out-of-plane for a) magic blue-, b) Mo(tfd-COCF₃)₃-, and c) F₄-TCNQ-doped WSe₂. Shaded red and blue correspond to the positions of WSe₂ layer and molecular dopants, respectively. Insets show isosurface plots of the charge density difference for molecular dopant binding site in WSe₂. Yellow and blue isosurfaces represent the excessive and deficient charge densities, respectively. The isosurface level is equal to 0.002 e Å⁻³. Computed total and partial density of states (DOS) of d) magic blue-, e) Mo(tfd-COCF₃)₃-, and f) F₄-TCNQ-doped WSe₂. The LUMO and HOMO levels of three molecular dopants are labeled. Since the total DOS of magic blue, Mo(tfd-COCF₃)₃, and F₄-TCNQ fifteen times larger than the actual values.

where $N_{\text{imp-doped}}$ and $N_{\text{imp-pristine}}$ are charged impurity density in doped and pristine WSe₂ devices, respectively.

The degree of Coulomb scattering induced by the dopant counterions will depend on the amount of charge transfer between WSe₂ and the dopant molecule. In molecular scale, the amount of charge transfer per dopant molecule (denoted as $\Delta Q = \alpha e$, where α is a constant that determines the effective degree of charge transfer) is dependent on the intrinsic molecular properties. In this case, the experimentally determined $\Delta N_{\rm imp}$ values should not be treated as the actual charged impurity density (i.e., $\Delta n_{imp} = n_{imp-doped} - n_{imp-pristine}$, see Equation (2)) generated by doping, but should be treated as an effective charged impurity density generated by doping since the scattering potential of each dopant molecule can differ by a factor of α from a unit charge (i.e., $\Delta N_{imp} = \Delta n_{imp}$ when $\alpha = 1$). For comparing the $\Delta N_{\rm imp}$ between dopants, we analyzed the relation between $\Delta N_{\rm imp}$ and $\Delta n_{\rm WSe2}$ to obtain the α value of each dopant corresponding to the transferred charge (more details can be found in Section S8, Supporting Information). Considering the dopant concentration and the value of α in combination with the theoretical model, we obtain a simple relation, $\Delta N_{\rm imp} = \alpha \Delta n_{\rm WSe2}$. In other words, the larger the value of α , the larger the contribution of each dopant molecule toward Coulomb scattering. Figure 4b shows the $\Delta N_{\rm imp}$ of each dopant as a function of Δn_{WSe2} , from which we found that ΔN_{imp} for Mo(tfd- $COCF_{3}$ is the largest, followed by magic blue and F_{4} -TCNQ at the same doping level. That is, α value for Mo(tfd-COCF₃)₃ ($\alpha = 0.64$) is the largest, followed by magic blue (0.38) and F₄-TCNQ (0.20), which follows the trend of the LUMO level position of dopant molecules (Mo(tfd-COCF₃)₃ (5.6 eV) \approx magic blue (5.5 eV) > F₄-TCNQ (5.2 eV)). However, the significantly lower α value of magic blue than that of Mo(tfd-COCF₃)₃ could not be explained solely by their LUMO levels. This hints toward other factors that need to be considered for predicting the α value of a dopant molecule. In addition to the Coulomb scattering, the electrical characteristics of the doped FET devices indicated the role of the dopant ions as generating trap states which are also related to the value of α (see Figure S10, Supporting Information). Thus, reducing charged impurity scattering by using a dopant with low α is desired for controlling the carrier density of WSe₂ while maintaining a low subthreshold swing value and high mobility.

To understand the microscopic nature of charge transfer process between WSe2 and the dopant molecules, the experimentally determined values of α for each dopant can be viewed against in-depth van der Waals (vdW)-corrected DFT calculations (details on DFT calculations are provided in Supporting Information). In principle, the charge transfer process at the WSe₂/molecule interface is a complex problem that depends on multiple variables such as the relative frontier orbital levels, available density of states, and dielectric properties, all of which are considered in our calculations. Although our GGA level of theory is subject to slight limitations (i.e., the exact position of the frontier orbital levels and dynamic dielectric properties),^[47–50] we can expect qualitative trends to be meaningful as shown in previous DFT studies^[25,40,51,52] and our approaches are still relevant. From the vdW-corrected DFT calculations, we obtained the planaraveraged charge density profiles of the charge density difference $\Delta \rho = \rho_{doped-WSe_2} - (\rho_{WSe_2} + \rho_{dopant})$ along the *c*-axis for magic blue, Mo(tfd-COCF₃)₃, and F₄-TCNQ as shown in Figure 5a-c. Using DVANCED

Table 1. Computed lowest unoccupied molecular orbitals (LUMOs) (in eV), the highest occupied molecular orbitals (HOMOs) (in eV), and the transferred charge per a dopant (in *e*) for WSe₂, F_4 -TCNQ, magic blue, and Mo(tfd-COCF₃)₃ compared to the experiments. The reference is the vacuum level (= 0 eV).

	LUMO		НОМО		Transferred charge	
	DFT	Exp.	DFT	Exp.	DFT	Exp.
WSe ₂	-3.9	_	-5.1	_	_	_
F ₄ -TCNQ	-6.0	-5.2	-7.3	-8.3	0.38	0.20
Magic blue	-5.9	-5.5	-6.2	_	0.24	0.38
Mo(tfd-COCF ₃) ₃	-5.8	-5.6	-7.1	-7.8	0.35	0.64

these profiles, we obtained the amount of charge transfer per dopant (ΔQ_{DFT}) for magic blue (0.24e), Mo(tfd-COCF₃)₃ (0.35e), and F₄-TCNQ (0.38e), respectively (details on the calculation in Supporting Information). We also performed layer number dependent ΔQ_{DFT} calculations and confirmed that the ΔQ_{DFT} values of each dopant tend to converge as the number of WSe₂ layers increases (see Figure S11, Supporting Information). The observed convergence could be correlated with the combined effect of available density of states in WSe2 (which increases as the number of layers increases) and screening effect (which saturates the doping effect above a certain layer thickness). In addition, in order to validate our DFT results, we conducted the thickness dependent low-temperature measurements with different doping concentrations of magic blue dopants as a representative (see Figure S12, Supporting Information). We confirmed that the trend of the extracted α value versus the number of layers is similar to that predicted by the DFT calculations (see Figure S13, Supporting Information). In overall, our DFT calculations show an unambiguous p-type doping behavior for all the dopants and the same trend in ΔQ_{DFT} for magic blue and Mo(tfd-COCF₃)₃ as expected from *a* values extracted from the experiments. The origin of the deviation of ΔQ_{DFT} from αe for F₄-TCNQ is elaborated in Section S15, Supporting Information.

We have then examined the electronic structures of three molecular dopants and doped-WSe2 systems to elucidate the relation between the electronic structure and the charge transfer. Figure 5d-f shows the computed density of states (DOS) of WSe2 (black line) with the dopant adsorptions (shaded black) and the partial DOS of magic blue, Mo(tfd-COCF₃)₃, and F₄-TCNQ (each atomic contribution represented in different colors). The computed DOS of the doped WSe₂ show clear evidence for p-type doping for all the dopants as shown from the downward shift in the Fermi level of WSe₂ (see Figure S14, Supporting Information for separate DOS of WSe₂ and the dopants) and partial filling of the DOS at LUMO for all the dopant molecules. More specifically, for magic blue, the DOS at LUMO level is mainly contributed by the atoms composing the tri-bromophenyl cation (C, N, and Br) which agrees well with the isosurface plot of the charge difference as visualized in the inset of Figure 5a (yellow region). In our DFT calculations, we found that the ground state of magic blue is the doublet state (S = 1/2) while that of the other two is the singlet state (S = 0). Thus, we plotted the DOS of magic blue with spin up and down components separately as shown in Figure 5d. It is worthy to note that ΔQ_{DFT} for magic blue (0.24e) is smaller than that of Mo(tfd-COCF₃)₃ (0.35e) even

though the LUMO level position and the dielectric constant of magic blue are comparable to those of $Mo(tfd-COCF_3)_3$ (see Table 1) which is similar to the trend in αe values (0.38e for magic blue and 0.64e for Mo(tfd-COCF₃)₃). According to our computed DOS (Figure 5d), the reduced spin degree of freedom in charge-transfer^[53] from WSe₂ to magic blue can be the origin of the smaller value of ΔQ_{DFT} of magic blue. Our computed DOS results clearly show the selective transfer of spin-down electrons from WSe2 to the LUMO of magic blue. In overall, our DFT calculation implies that spin configuration as well as the intrinsic frontier orbital positions and dielectric constants of molecules play a significant role in determining the amount of charge transfer between WSe₂ and dopants, which broadens the scope of molecular properties that need to be considered for predicting the charge transfer effect. In addition to the well-established energetic role of frontier orbital position of dopants, we found that the other factors, such as spin configuration and dielectric constant of molecules, should also be considered to predict the charge transfer effect. Our study provides some key insights toward understanding the microscopic doping phenomenon and directly relating this to macroscopic device studies.

3. Conclusion

In conclusion, we explored dopant dependent charge transport of molecular-doped WSe2 FETs for three different dopants via experimental determination of the intrinsic mobility values of the doped WSe₂ channels. The reduction in the intrinsic mobility after doping could be interpreted as the creation of charged impurities in the form of dopant counterions on the WSe₂ surface. Comparison between the three dopants showed that the larger the degree of charge transfer between WSe₂ and dopants, the larger the carrier density generated in WSe₂, but also the larger the degree of Coulomb scattering induced by each dopant counterion. An agreement with DFT calculations corroborated the importance of considering the selective spin transfer as well as the relative frontier orbital positions of the dopant molecules. Our results will pave the way for controlling the carrier density and conductivity levels of TMDC through molecular doping and provide criteria for selecting appropriate molecular dopants for doping applications in TMDC nanoelectronic devices such as carrier-type control of the active channel and contact doping for low-power operation.

ADVANCED SCIENCE NEWS

www.advancedsciencenews.com

4. Experimental Section

Fabrication and Electrical Characterization of Tungsten Diselenide Field-Effect Transistor Devices: WSe2 was mechanically exfoliated from a bulk WSe_2 crystal and transferred to $SiO_2/p++$ Si substrate. The suitable WSe_2 flakes were founded using an optical microscope and the thickness of the flakes was measured by an AFM system (NX 10 AFM, Park Systems). After double electron resist layers, methyl methacrylate/poly(methyl methacrylate), were spin-coated, the source-drain electrodes patterns were made by using an electron-beam lithography system (JSM-6510, JEOL). Subsequently, Ti (2 nm)/Pd (40 nm)/Au (20 nm) layers were deposited by using an electron-beam evaporator (KVE-2004L, Korea Vacuum Tech.). Here, Pd was used to enhance hole injection and p-type conduction of WSe₂ FET due to its high work function (\approx 5.6 eV). And, Ti was used for the adhesive layer to SiO₂ and Au was used for compatibility with wire bonding. The fabricated WSe2 FET devices were annealed at 200 $\,^\circ\text{C}$ for 1 h before the electrical measurement. The electrical characteristics of the WSe2 FETs were measured through a semiconductor parameter analyzer (Keithley 4200-SCS) and a cryostat system (CS204*I-FMX-12, Advanced Research Systems). In device fabrication, WSe₂ flakes with a layer thickness from 6 to 15 nm were used, confirmed by AFM. In this thickness range, the substrate effect could be minimized due to a larger thickness relative to the expected screening length of WSe2.^[54] Also, it was confirmed that the effect of the thickness of WSe2 on doping was not dominant in this thickness range (see Figures S1 and S9 and Table S5, Supporting Information).

Doping Treatment: To make a doping solution, F_4 -TCNQ and Mo(tdf-COCF₃)₃ were dissolved in butyl acetate and magic blue was dissolved in dichloromethane. About 20 μ L of the molecular dopant solution was used in a drop-casting process for each dopant.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements

J.-K.K. and K.C. contributed equally to this work. The authors appreciate the financial support of the National Research Foundation of Korea (NRF) grant (Nos. 2021R1A2C3004783 and NRF-2021R1C1C1010266) and the Nano Material Technology Development Program grant (No. 2021M3H4A1A02049651) through NRF funded by the Ministry of Science and ICT (MSIT) of Korea. K.C. appreciates the financail support of the NRF grant funded by the Korea government (MSIT) (No. 2021R1C1C2091728). J.-H.L.'s work was supported by the KIST Institutional Program (Project No. 2E31201). Computational resources provided by KISTI Supercomputing Centre (Project No. KSC-2020-CRE-0189) are gratefully acknowledged.

Conflict of Interest

The authors declare no conflict of interest.

Data Availability Statement

Research data are not shared.

Keywords

charged impurity scattering, density functional theory, field effect transistors, surface charge transfer doping, tungsten diselenide



www.advmat.de

Received: February 26, 2021 Revised: August 15, 2021 Published online: September 17, 2021

- [1] H. Li, J.-K. Huang, Y. Shi, L.-J. Li, Adv. Mater. Interfaces 2019, 6, 1900220.
- [2] M. Chhowalla, D. Jena, H. Zhang, Nat. Rev. Mater. 2016, 1, 16052.
- [3] B. Liu, A. Abbas, C. Zhou, Adv. Electron. Mater. 2017, 3, 1700045.
- [4] K. Cho, J. Pak, S. Chung, T. Lee, ACS Nano 2019, 13, 9713.
- [5] A. Nourbakhsh, A. Zubair, R. N. Sajjad, A. Tavakkoli K G, W. Chen, S. Fang, X. Ling, J. Kong, M. S. Dresselhaus, E. Kaxiras, K. K. Berggren, D. Antoniadis, T. Palacios, *Nano Lett.* **2016**, *16*, 7798.
- [6] S. B. Desai, S. R. Madhvapathy, A. B. Sachid, J. P. Llinas, Q. Wang, G. H. Ahn, G. Pitner, M. J. Kim, J. Bokor, C. Hu, H.-S. P. Wong, A. Javey, *Science* **2016**, *354*, 99.
- [7] G. M. Marega, Y. Zhao, A. Avsar, Z. Wang, M. Tripathi, A. Radenovic, A. Kis, *Nature* **2020**, *587*, 72.
- [8] D. Y. Qiu, F. H. da Jornada, S. G. Louie, Phys. Rev. Lett. 2013, 111, 216805.
- [9] D. Y. Qiu, F. H. da Jornada, S. G. Louie, Phys. Rev. B 2016, 93, 235435.
- [10] H.-S. Lee, V. K. Sangwan, W. A. G. Rojas, H. Bergeron, H. Y. Jeong, J. Yuan, K. Su, M. C. Hersam, *Adv. Funct. Mater.* **2020**, *30*, 2003683.
- [11] A. B. Sachid, M. Tosun, S. B. Desai, C.-Y. Hsu, D.-H. Lien, S. R. Madhvapathy, Y.-Z. Chen, M. Hettick, J. S. Kang, Y. Zeng, J.-H. He, E. Y. Chang, Y.-L. Chueh, A. Javey, C. Hu, *Adv. Mater.* **2016**, *28*, 2547.
- [12] A. Avsar, K. Marinov, E. G. Marin, G. lannaccone, K. Watanabe, T. Taniguchi, G. Fiori, A. Kis, *Adv. Mater.* **2018**, *30*, 1707200.
- [13] Q. Liang, Q. Wang, Q. Zhang, J. Wei, S. X. Lim, R. Zhu, J. Hu, W. Wei, C. Lee, C. Sow, W. Zhang, A. T. S. Wee, *Adv. Mater.* **2019**, *31*, 1807609.
- [14] T. Zhang, K. Fujisawa, F. Zhang, M. Liu, M. C. Lucking, R. N. Gontijo, Y. Lei, H. Liu, K. Crust, T. Granzier-Nakajima, H. Terrones, A. L. Elías, M. Terrones, ACS Nano 2020, 14, 4326.
- [15] H. Gao, J. Suh, M. C. Cao, A. Y. Joe, F. Mujid, K.-H. Lee, S. Xie, P. Poddar, J.-U. Lee, K. Kang, P. Kim, D. A. Muller, J. Park, *Nano Lett.* **2020**, *20*, 4095.
- [16] S. K. Pandey, H. Alsalman, J. G. Azadani, N. Izquierdo, T. Low, S. A. Campbell, *Nanoscale* **2018**, *10*, 21374.
- [17] D. Kiriya, M. Tosun, P. Zhao, J. S. Kang, A. Javey, J. Am. Chem. Soc. 2014, 136, 7853.
- [18] H. Fang, M. Tosun, G. Seol, T. C. Chang, K. Takei, J. Guo, A. Javey, Nano Lett. 2013, 13, 1991.
- [19] D.-H. Kang, J. Shim, S. K. Jang, J. Jeon, M. H. Jeon, G. Y. Yeom, W.-S. Jung, Y. H. Jang, S. Lee, J.-H. Park, ACS Nano 2015, 9, 1099.
- [20] M.-Y. Tsai, S. Zhang, P. M. Campbell, R. R. Dasari, X. Ba, A. Tarasov, S. Graham, S. Barlow, S. R. Marder, E. M. Vogel, *Chem. Mater.* **2017**, *29*, 7296.
- [21] S. Zhang, H. M. Hill, K. Moudgil, C. A. Richter, A. R. H. Walker, S. Barlow, S. R. Marder, C. A. Hacker, S. J. Pookpanratana, *Adv. Mater.* 2018, *30*, 1802991.
- [22] W. Chen, S. Chen, D. Qi, X. Gao, A. Wee, J. Am. Chem. Soc. 2007, 129, 10418.
- [23] W. Chen, D. Qi, X. Gao, A. Wee, Prog. Surf. Sci. 2009, 84, 279.
- [24] Y. Zhao, K. Xu, F. Pan, C. Zhou, F. Zhou, Y. Chai, Adv. Funct. Mater. 2017, 27, 1603484.
- [25] M. Gobbi, E. Orgiu, P. Samorì, Adv. Mater. 2018, 30, 1706103.
- [26] J. Wang, Z. Ji, G. Yang, X. Chuai, F. Liu, Z. Zhou, C. Lu, W. Wei, X. Shi, J. Niu, L. Wang, H. Wang, J. Chen, N. Lu, C. Jiang, L. Li, M. Liu, Adv. Funct. Mater. 2018, 28, 1806244.
- [27] L. Yu, A. Zubair, E. J. G. Santos, X. Zhang, Y. Lin, Y. Zhang, T. Palacios, *Nano Lett.* **2015**, *15*, 4928.

ADVANCED SCIENCE NEWS

www.advancedsciencenews.com



www.advmat.de

- [28] H. G. Ji, P. Solís-Fernández, D. Yoshimura, M. Maruyama, T. Endo, Y. Miyata, S. Okada, H. Ago, *Adv. Mater.* **2019**, *31*, 1903613.
- [29] F. Schedin, A. K. Geim, S. V. Morozov, E. W. Hill, P. Blake, M. I. Katsnelson, K. S. Novoselov, *Nat. Mater.* 2007, *6*, 652.
- [30] N. Ma, D. Jena, Phys. Rev. X 2014, 4, 011043.
- [31] Z.-Y. Ong, M. V. Fischetti, Phys. Rev. B 2013, 88, 165316.
- [32] C. M. Smyth, R. Addou, S. McDonnell, C. L. Hinkle, R. M. Wallace, 2D Mater. 2017, 4, 025084.
- [33] F. A. Rasmussen, K. S. Thygesen, J. Phys. Chem. C 2015, 119, 13169.
- [34] S. A. Paniagua, J. Baltazar, H. Sojoudi, S. K. Mohapatra, S. Zhang, C. L. Henderson, S. Graham, S. Barlow, S. R. Marder, *Mater. Horiz.* 2014, 1, 111.
- [35] K. Kang, S. Watanabe, K. Broch, A. Sepe, A. Brown, I. Nasrallah, M. Nikolka, Z. Fei, M. Heeney, D. Matsumoto, K. Marumoto, H. Tanaka, S.-i. Kuroda, H. Sirringhaus, *Nat. Mater.* **2016**, *15*, 896.
- [36] Z. Wang, Q. Li, Y. Chen, B. Cui, Y. Li, F. Besenbacher, M. Dong, NPG Asia Mater 2018, 10, 703.
- [37] A. Allain, A. Kis, ACS Nano 2014, 8, 7180.
- [38] Y. L. Huang, Y. J. Zheng, Z. Song, D. Chi, A. T. S. Wee, S. Y. Quek, *Chem. Soc. Rev.* 2018, 47, 3241.
- [39] Z. Song, Q. Wang, M.-Y. Li, L.-J. Li, Y. J. Zheng, Z. Wang, T. Lin, D. Chi, Z. Ding, Y. L. Huang, A. T. S. Wee, *Phys. Rev. B* 2018, *97*, 134102.
- [40] M. Yarali, Y. Zhong, S. N. Reed, J. Wang, K. A. Ulman, D. J. Charboneau, J. B. Curley, D. J. Hynek, J. V. Pondick, S. Yazdani, N. Hazari, S. Y. Quek, H. Wang, J. J. Cha, *Adv. Electron. Mater.* **2021**, 7, 2000873.

- [41] X. Cui, G.-H. Lee, Y. D. Kim, G. Arefe, P. Y. Huang, C.-H. Lee, D. A. Chenet, X. Zhang, L. Wang, F. Ye, F. Pizzocchero, B. S. Jessen, K. Watanabe, T. Taniguchi, D. A. Muller, T. Low, P. Kim, J. Hone, *Nat. Nanotechnol.* **2015**, *10*, 534.
- [42] M.-K. Joo, B. H. Moon, H. Ji, G. H. Han, H. Kim, G. Lee, S. C. Lim, D. Suh, Y. H. Lee, *Nano Lett.* **2016**, *16*, 6383.
- [43] A. Rai, A. Valsaraj, H. C. P. Movva, A. Roy, R. Ghosh, S. Sonde, S. Kang, J. Chang, T. Trivedi, R. Dey, S. Guchhait, S. Larentis, L. F. Register, E. Tutuc, S. K. Banerjee, *Nano Lett.* **2015**, *15*, 4329.
- [44] H. Ehrenreich, Phys. Rev. 1960, 120, 1951.
- [45] A. Gold, Appl. Phys. Lett. 2010, 96, 242111.
- [46] M. P. Persson, H. Mera, Y.-M. Niquet, C. Delerue, M. Diarra, *Phys. Rev. B* 2010, *82*, 115318.
- [47] J. P. Perdew, A. Zunger, Phys. Rev. B 1981, 23, 5048.
- [48] J. P. Perdew, R. G. Parr, M. Levy, J. L. Balduz Jr., Phys. Rev. Lett. 1982, 49, 1691.
- [49] J. P. Perdew, M. Levy, Phys. Rev. Lett. 1983, 51, 1884.
- [50] L. J. Sham, M. Schlüter, Phys. Rev. Lett. 1983, 51, 1888.
- [51] A. Kumar, K. Banerjee, M. Dvorak, F. Schulz, A. Harju, P. Rinke, P. Liljeroth, ACS Nano 2017, 11, 4960.
- [52] H.-Z. Tsai, A. A. Omrani, S. Coh, H. Oh, S. Wickenburg, Y.-W. Son, D. Wong, A. Riss, H. S. Jung, G. D. Nguyen, G. F. Rodgers, A. S. Aikawa, T. Taniguchi, K. Watanabe, A. Zettl, S. G. Louie, J. Lu, M. L. Cohen, M. F. Crommie, ACS Nano 2015, 9, 12168.
- [53] T. R. Kafle, B. Kattel, S. D. Lane, T. Wang, H. Zhao, W.-L. Chan, ACS Nano 2017, 11, 10184.
- [54] V. K. Sangwan, M. C. Hersam, Annu. Rev. Phys. Chem. 2018, 69, 299.